

## Article

# Machine Learning Aided Prediction of Glass-Forming Ability of Metallic Glass

Chengcheng Liu <sup>1,2</sup>, Xuandong Wang <sup>2</sup>, Weidong Cai <sup>2</sup>, Yazhou He <sup>2</sup> and Hang Su <sup>2,\*</sup>

<sup>1</sup> Institute of Structural Steel, Central Iron and Steel Research Institute, Beijing 100081, China; ustbliuchengcheng@foxmail.com

<sup>2</sup> Material Digital R&D Center, China Iron and Steel Research Institute Group, Beijing 100081, China; tony775316@163.com (X.W.); gsgxwcd@163.com (W.C.); windyhermo@163.com (Y.H.)

\* Correspondence: hangsu@vip.sina.com

**Abstract:** The prediction of the glass-forming ability (GFA) of metallic glasses (MGs) can accelerate the efficiency of their development. In this paper, a dataset was constructed using experimental data collected from the literature and books, and a machine learning-based predictive model was established to predict the GFA. Firstly, a classification model based on the size of the critical diameter ( $D_{\max}$ ) was established to determine whether an alloy system could form a glass state, with an accuracy rating of 0.98. Then, regression models were established to predict the crystallization temperature ( $T_x$ ), glass transition temperature ( $T_g$ ), and liquidus temperature ( $T_l$ ) of MGs. The  $R^2$  of the prediction model obtained in the test set was greater than 0.89, which showed that the model had good prediction accuracy. The key features used by the regression models were analyzed using variance, correlation, embedding, recursive, and exhaustive methods to select the most important features. Furthermore, to improve the interpretability of the prediction model, feature importance, partial dependence plot (PDP), and individual conditional expectation (ICE) methods were used for visualization analysis, demonstrating how features affect the target variables. Finally, taking Zr-Cu-Ni-Al system MGs as an example, a prediction model was established using a genetic algorithm to optimize the alloy composition for high GFA in the compositional space, achieving the optimal design of alloy composition.

**Keywords:** metallic glass; glass-forming ability; machine learning; optimal design



**Citation:** Liu, C.; Wang, X.; Cai, W.; He, Y.; Su, H. Machine Learning Aided Prediction of Glass-Forming Ability of Metallic Glass. *Processes* **2023**, *11*, 2806. <https://doi.org/10.3390/pr11092806>

Academic Editors: Wei Lv, Wanqiang Liu, Li Yang, Weisen Zheng and Feng Wang

Received: 23 August 2023  
Revised: 7 September 2023  
Accepted: 9 September 2023  
Published: 21 September 2023



**Copyright:** © 2023 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

## 1. Introduction

Solid-state structure can be categorized into crystalline, quasi-crystalline, and amorphous states based on the arrangement of particles. Crystalline structures exhibit a regular and ordered arrangement of particles in three-dimensional space, demonstrating long-range order and translation symmetry. These structures possess lower energy and higher stability, which explains why most everyday solid materials showcase crystalline structures. Quasi-crystalline structures, on the other hand, possess long-range orientational order but lack long-range translational order. Amorphous structures, characterized by a lack of ordered particle arrangements, do not exhibit periodicity or translational symmetry. Although they exhibit short-range order, they display long-range disorder. Metallic glass (MGs) serves as a typical example of amorphous materials. Due to the unique particle arrangement in MGs, which is less susceptible to lattice defects, they possess exceptional properties that are not commonly found in crystalline structures. These properties include high strength [1], high wear resistance [2], high elastic limits [3], corrosion resistance [4], and excellent soft magnetic properties [5]. Consequently, MGs hold vast potential for numerous applications. However, the glass-forming ability (GFA) of MGs plays a crucial role in limiting their application as conventional materials. The GFA refers to a metal's ability to form an amorphous glassy state during the cooling process. A higher GFA indicates a

greater ease of achieving the amorphous state, resulting in improved processability and performance enhancements.

The prediction of GFA plays a crucial role in understanding the behavior of MGs, particularly in investigating the impact of composition on the GFA. Traditionally, research on GFA has primarily focused on experimentation, where the GFA is obtained by adjusting the composition of MGs. This approach heavily relies on empirical knowledge and experience. However, the process of finding MGs with high GFA through trial and error can be time-consuming and resource intensive. To address this challenge, machine learning approaches have gained significant attention in the field of GFA. By leveraging large amounts of experimental data, machine learning techniques have been applied to analyze and predict the GFA of MGs. These approaches enable researchers to develop models that can accurately predict the GFA of new compositions, thereby reducing the need for extensive experimentation. Currently, most GFA predictions are based on estimating the critical diameter ( $D_{\max}$ ) [6–13]. While  $D_{\max}$  provides a direct measure of GFA, it is influenced by various factors, including fabrication processes, sample shape, and testing methods. Consequently, even MGs with identical compositions can exhibit significant variations in  $D_{\max}$ , resulting in low prediction accuracy when developing models. This limitation impedes the wider application of machine learning in the field of metallic glasses. Researchers have discovered strong correlations between the GFA and the crystallization temperature ( $T_x$ ), glass transition temperature ( $T_g$ ), and liquidus temperature ( $T_l$ ) of MGs, which can be reliably measured using thermal analysis techniques.  $T_x$  refers to the temperature at which a material transitions from a solid or amorphous state to a crystalline state.  $T_g$  represents the temperature below which an amorphous material, such as glass or a polymer, becomes rigid or solid-like. This is the point where the material transitions from a supercooled liquid to a glassy state.  $T_l$  refers to the temperature at which a substance completely melts and turns into a liquid phase. It is the highest temperature at which both solid and liquid phases coexist in equilibrium. Therefore, it is reasonable to explore alternative parameters that can capture GFA as a composite of these temperature-related properties, thus overcoming the limitations associated with  $D_{\max}$ . Turnbull et al. [14] introduced the reduced glass transition temperature ( $T_{rg} = T_g/T_l$ ) based on the crystallization kinetics of supercooled liquids as a way to characterize GFA. Higher  $T_{rg}$  values are indicative of greater GFA. Chen et al. [15] and Inoue et al. [16] proposed that the supercooled liquid region ( $\Delta T_x = T_x - T_g$ ) of MGs reflects the stability of the supercooled liquid, and can thus serve as a measure of the GFA. Lu et al. [17] combined  $T_{rg}$  and  $\Delta T_x$  to introduce a novel GFA parameter  $\gamma$ , where  $\gamma = T_x/(T_g + T_l)$ . Experimental results have demonstrated that  $\gamma$  is an effective indicator of the GFA.

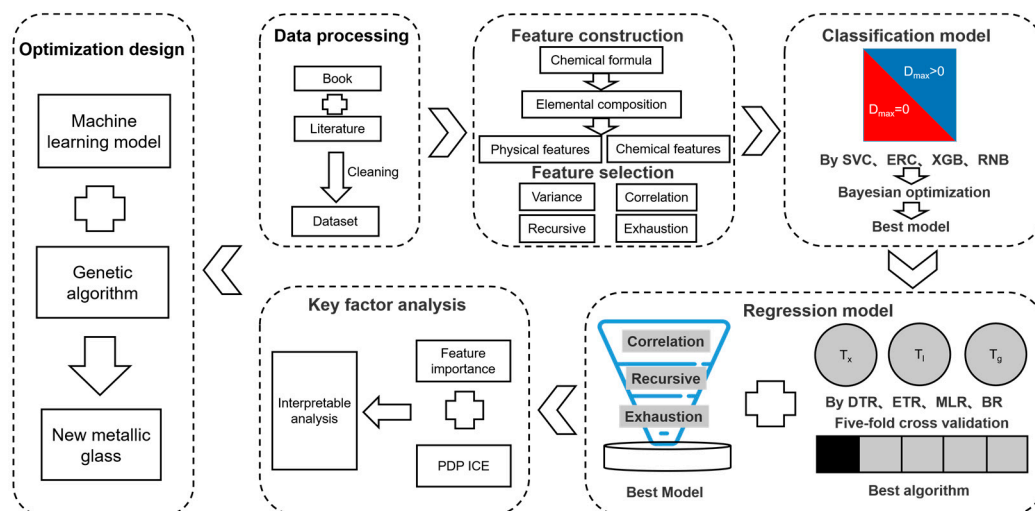
These studies suggest that parameters like  $T_x$ ,  $T_g$ , and  $T_l$  are effective indicators of the GFA of MGs. However, there is currently limited research focused on predicting these temperature parameters. In this study, we gathered a substantial amount of the GFA data for MGs from literature sources and utilized machine learning techniques to develop predictive models. Furthermore, we conducted feature selection and interpretability analysis to identify the key factors influencing these parameters. We proposed composition strategies to enhance the GFA of MGs and combined them with a genetic algorithm to explore novel MGs. These findings provide valuable insights for the future design of MGs.

## 2. Materials and Methods

### 2.1. Machine Learning Model Framework

In this study, a machine learning-based prediction model was developed to estimate the GFA of MGs. This study aimed to investigate the impact of key feature variables on GFA, providing theoretical and model support for optimizing and designing MG compositions. The research process followed a specific framework, as illustrated in Figure 1. The framework involved several steps, including data processing, feature construction, construction of classification and regression models, analysis of key factors, and optimization design. During the data processing phase, a comprehensive dataset was compiled by collecting

$D_{\max}$ ,  $T_x$ ,  $T_g$ , and  $T_l$  values of different alloy compositions from the literature and books. The chemical formulas of the alloys were then transformed into machine-learning-friendly features through careful feature construction. These transformed features were utilized to establish a classification model to determine whether  $D_{\max}$  was greater than 0. For alloys with  $D_{\max}$  greater than 0, separate regression models were developed to predict  $T_x$ ,  $T_g$ , and  $T_l$ . To further refine the models, feature selection techniques were employed to simplify and optimize their performance. Finally, a synergistic combination of the predictive models and a genetic algorithm was utilized to explore the compositional space and identify novel alloys with high GFA.



**Figure 1.** Process framework for establishing GFA prediction model based on machine learning.

## 2.2. Dataset Establishment

The dataset used in this study was primarily compiled by Ward et al. [18] and included experimental data on the GFA of MGs up until 2018. The data were sourced from books [19] and the literature [20–29]. Subsequently, additional experimental data published from 2018 to the present [10,12,30–65] were collected and incorporated into the dataset. This combination of sources formed the basis of the new dataset for this study. Figure 2 shows the distribution of elements contained in the data set in the periodic table of elements. The dataset primarily consists of the chemical formulae of MGs and their corresponding  $D_{\max}$ ,  $T_l$ ,  $T_x$ , and  $T_g$ . The sample quantities for  $D_{\max}$ ,  $T_l$ ,  $T_x$ , and  $T_g$  are 10,194, 751, 791, and 846, respectively.

## 2.3. Feature Construction and Selection

Given that the dataset solely considers the impact of chemical formulas on the GFA, the process of feature construction primarily revolves around the composition of MGs. Firstly, the chemical formulas are transformed into elemental compositions. Subsequently, the physical and chemical properties of each element are taken into consideration, and a weighted average is computed based on the elemental composition, as shown in Table 1. Moreover, in order to enhance the dimensionality of the features and mitigate overfitting [66], statistical features, such as the maximum value, minimum value, and absolute deviation of elemental properties, are incorporated. This results in the generation of 132 descriptors, which collectively form the input data for forecasting the GFA.

To identify the key features that influence the GFA of MGs, it is essential to conduct feature selection on the constructed dataset. Firstly, a variance filter is applied to identify and eliminate features with low variances. Subsequently, F-tests [67] are utilized to detect and remove highly correlated features. Additionally, an embedding method [68] is employed to further refine the feature selection process. This method evaluates the contribution of each feature to the model by assigning weights during training and automatically eliminates

features with low importance below a predefined threshold. Following the embedding method, a recursive feature elimination [69] technique is used to iteratively remove features. The criterion of threshold selection in the embedding method is to increase the threshold and reduce the number of features of the model without affecting the accuracy of the model as much as possible. The recursive method constructs different feature subsets to train the model, uses *MAE* as the evaluation index of the model, and gradually selects the features that are beneficial to the model performance, thus realizing the feature selection. The criterion of feature number selection in recursion is that when the number of features continues to increase, the accuracy of the model does not improve, and it is considered that this is the best feature number. This iterative process effectively reduces the dimensionality of the features. Finally, an exhaustive search [70] is conducted on the remaining features to identify the key features that significantly influence the GFA. The exhaustive method is to combine the features selected by the recursive method, select  $R^2$  and *MAE* as evaluation indexes, calculate the performance of the model under various combinations, and obtain the best performance feature combination.

The figure shows a periodic table of elements with a color-coding scheme. Elements are colored in red, green, blue, and white. A diagram above the table shows a cross shape with a central white square and four colored squares (red, green, blue, white) extending from it, labeled  $D_{max}$ . To the right of the diagram are labels  $T_l$ ,  $T_x$ , and  $T_g$ . The periodic table includes elements from Hydrogen (H) to Oganesson (Og), with Lanthanides and Actinides shown in separate rows at the bottom.

**Figure 2.** Types of elements contained in datasets. \* Lanthanides. \*\* Actinides.

**Table 1.** Feature name and description.

Feature Name	Description	Feature Name	Description
Number	Atomic number	N(s,p,d,f) valence	Number of electrons in (s,p,d,f) level
Mendeleev number	Mendeleev number	N(s,p,d,f) unfilled	Number of unfilled electrons in (s,p,d,f) energy level
Atomic weight	Atomic weight	N unfilled	Number of unfilled energy layer electrons
Melting temperature	Melting point	GS volume_pa	Average volume of atoms
Column	Column of the element	GS bandgap	Band gap of elements
Row	Row of the element	GS magmom	Magnetic moment of element
Covalent radius	Covalent bond radius	Space group number	Group serial number
Electronegativity	Electronegativity of elements	N valence	Energy layer electron number

#### 2.4. Machine Learning Model and Feature Analysis

The construction of the machine learning model involves two main steps. Firstly, the classification stage uses  $D_{max} > 0$  or  $D_{max} = 0$  as the criterion, and commonly used

classification algorithms such as Support Vector Machine (SVC), Extremely Randomized Trees (ETC), XGBoost (XGC), and Naive Bayes (GNB) are applied to determine whether an alloy's composition can be transformed into Bulk Metallic Glasses (BMGs). Then, in the regression stage, regression algorithms, including Decision Tree Regression (DTR), Extremely Randomized Trees Regression (ETR), Multi-layer Perceptron Regression (MLP), and Linear Regression (LR), are employed to analyze the characterization parameters of BMGs' GFA. This aims to establish a predictive model that can estimate the relationship between composition and GFA. The data set is divided into the training set and test set according to 8:2. The training set is used to train the model, and the subsequent test set is used to verify the generalization ability of the model. In the comparison of regression algorithms, in order to avoid the problem of under-fitting or over-fitting models caused by randomly dividing training data, K-fold cross-validation is used to calculate the evaluation indexes of different models [71,72]. K-fold cross-validation involves dividing the dataset into k subsets, using k-1 subsets for training and one subset for testing in each iteration. The model is then run k times, and the average of the k results is considered the final evaluation metric. In this study, a five-fold cross-validation approach was adopted to ensure reliable model performance assessment.

For different machine learning models, classification and regression tasks require the use of different evaluation metrics. In the case of classification models, particularly binary classification models, the performance is typically assessed using a confusion matrix. The values within the confusion matrix [73] represent the model's classification predictions and include True Positive (TP), False Positive (FP), True Negative (TN), and False Negative (FN), as shown in Table 2. These values enable the calculation of various evaluation metrics, including Accuracy, Precision, Recall, and F1 score. The formulas for these metrics are presented in Equations (1)–(4).

$$\text{Accuracy} = \frac{TP + TN}{TP + FN + FP + TN}. \quad (1)$$

$$\text{Precision} = \frac{TP}{TP + FP} \quad (2)$$

$$\text{Recall} = \frac{TP}{TP + FN} \quad (3)$$

$$\text{F1} = \frac{2 \times \text{Precision} \times \text{Recall}}{\text{Precision} + \text{Recall}} \quad (4)$$

**Table 2.** The confusion matrix of binary classification.

Actual	Predicted	
	Positive	Negative
Positive	TP	FN
Negative	FP	TN

In order to further assess the effectiveness of different classification models, the Receiver Operating Characteristic (ROC) curve was employed to compare their generalization ability. The ROC curve illustrates the relationship between the TP and the FP of a classifier. In the ROC curve, the FP is plotted on the horizontal axis, and the TP is plotted on the vertical axis. Each point on the ROC curve corresponds to the TP and FP values obtained at a specific threshold used for classification. A classifier with a ROC curve closer to the top-left corner demonstrates better performance. Moreover, the area under the ROC curve serves as an indication of the model's performance, with a larger area suggesting better performance.

For regression models, evaluating the performance involves comparing the predicted values with the actual values. As such, Mean Squared Error (*MSE*), Mean Absolute Error (*MAE*), and Coefficient of Determination (*R*-squared,  $R^2$ ) serve as common evaluation metrics to assess the accuracy of the model. The formulas for these metrics are presented in Equations (5)–(7).

$$MSE = \sum_{i=1}^n \frac{1}{n} (\hat{y}_i - y_i)^2. \quad (5)$$

$$MAE = \sum_{i=1}^n \frac{1}{n} |\hat{y}_i - y_i| \quad (6)$$

$$R^2 = 1 - \frac{\sum_{i=1}^n (\hat{y}_i - y_i)^2}{\sum_{i=1}^n (\bar{y}_i - y_i)^2} \quad (7)$$

where  $y_i$  represents the actual value,  $\bar{y}_i$  represents the average of the actual values, and  $\hat{y}_i$  represents the predicted value. *MAE* and *MSE* measure the gap between the predicted and actual values, with a smaller value indicating smaller prediction errors. The  $R^2$  quantifies the level of fit between the predicted and actual values. A value closer to 1 implies a better fit between the predictions and the actual values.

Most machine learning models are often considered black box models, as they can establish the mapping between input features and output results without providing explicit expressions. This lack of interpretability is a common issue. To address this, the current paper introduces Partial Dependence Plots (PDP) and Individual Conditional Expectation Plots (ICE) [74–77]. The fundamental concept underlying PDP is to assess the impact of a single feature on the output while keeping the other features constant. This analysis helps reveal the nonlinear relationships present in the model's output. PDP is based on a straightforward observation: in many cases, the model's output depends not solely on the value of an individual variable but rather on the simultaneous changes in the combination of different variables. By simplifying this multivariate relationship, PDP can effectively describe the effect of a single feature on the model's output. Conversely, ICE plots are primarily employed to illustrate the relationship between the variable values of individual samples and their corresponding response values. ICE plots effectively reveal the nonlinear relationships and trends within the model's output. The calculation of ICE plots involves determining the changes in the model's predicted output for each individual sample as the feature values vary. The  $x$ -axis represents the arrangement of individual variable values, while the  $y$ -axis represents the predicted output for each sample. For each feature value, ICE plots visually capture the nonlinear changes in the specific feature by plotting the response values for each individual sample within a single graph.

## 2.5. Composition Optimization

Due to the huge search space, it is unrealistic to use enumeration to find the best point, so genetic algorithm is used to optimize the composition. NSGA-II [78,79] is a classic multi-objective optimization algorithm based on genetic algorithms. NSGA-II evaluates and selects individuals using non-dominated sorting and crowding distance to solve multi-objective optimization problems. The basic principles of the NSGA-II algorithm are as follows: Non-dominated Sorting, Crowding Distance Assignment, Selection, Crossover, Mutation, and Termination. The NSGA-II algorithm combines non-dominated sorting and crowding distance to find a set of non-dominated solutions in multi-objective optimization problems while maintaining the diversity and convergence of the population. This algorithm has the ability to maintain a diverse set of solutions and explore the solution space effectively. The non-dominated sorting ensures that the algorithm can find a balance between different objectives, rather than just searching for a single optimal solution. The crowding distance is used to measure the density of individuals within each non-dominated

level, reflecting the distribution of individuals in the objective space. By considering both non-dominated sorting and crowding distance, the NSGA-II algorithm is able to maintain a diverse population and avoid converging to local optima. This algorithm has been widely used and was proven to be effective in various applications.

### 3. Results

#### 3.1. Classification Model Evaluation and Feature Selection

Figure 3 shows the performance comparison of four classification models. Figure 3a shows a comparative analysis of four classification models in terms of performance metrics. The ETR and XGC models demonstrate superior performance compared to the SVC and GNB models. Both the ETR and XGC models exhibit accuracy, precision, recall, and F1-score values above 0.98, indicating their high accuracy in classifying various labels. Furthermore, Figure 3b shows an examination of the ROC curves for the four models, revealing that the ETR and XGC models yield curves positioned more towards the upper-left corner, encompassing larger areas. The ETR model has an enclosed area of 0.99, while the XGC model achieves a perfect score of 1. This signifies the superior performance of both models compared to XGC. Consequently, the XGC model was selected for the subsequent development of the classification model.

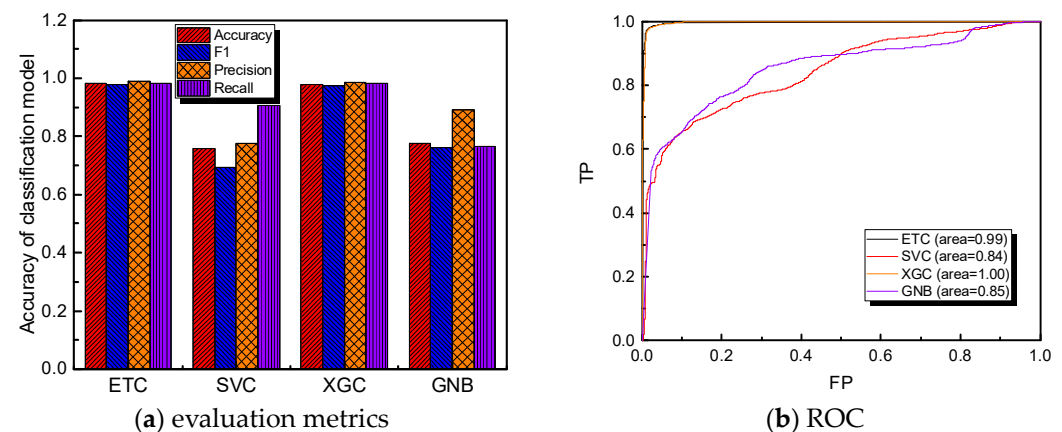


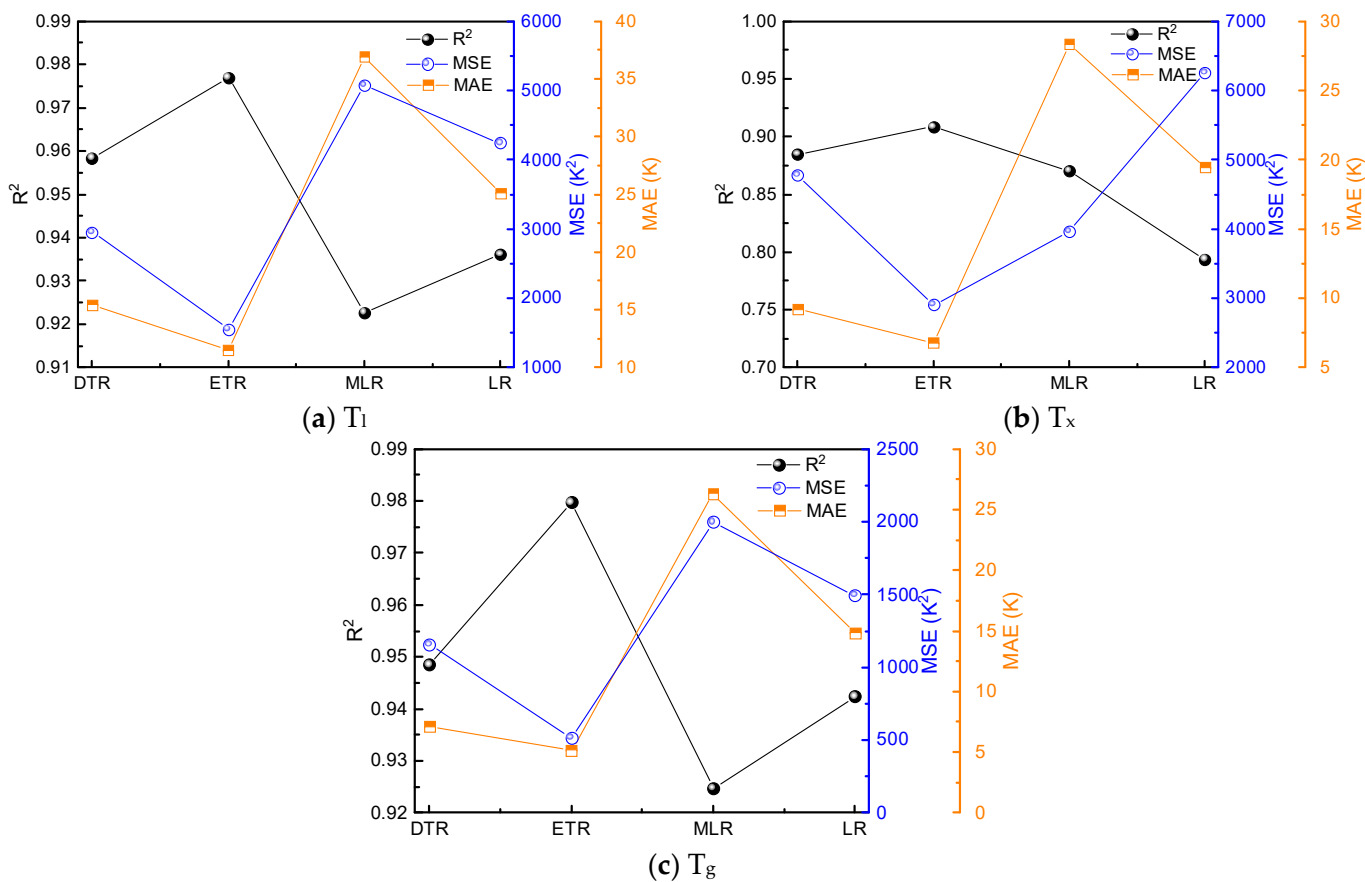
Figure 3. Comparison of four classification models.

#### 3.2. Regression Model Evaluation and Feature Selection

This study investigated the GFA of BMGs using three characterization parameters:  $T_l$ ,  $T_g$ , and  $T_x$ . Multiple machine learning algorithms were applied to predict these parameters, and a five-fold cross-validation method, similar to the previous evaluation, was employed for performance assessment. Figure 4 shows the performance of  $T_l$ ,  $T_g$ , and  $T_x$  on different regression models. It can be observed that among the four regression models, the ETR model exhibits the highest  $R^2$  and the lowest  $MSE$  and  $MAE$ . Overall, the results indicate that the ETR model demonstrates the best performance, and therefore, it was selected for subsequent predictions. Figure 5 compares the predicted and experimental values of  $T_l$ ,  $T_g$ , and  $T_x$  using the ETR model. The predicted values closely align with the diagonal line, indicating excellent accuracy of the model predictions. At the test set: the  $R^2$  of  $T_l$  reaches 0.98, with a  $MAE$  of 23.21 K and a  $MSE$  of 1481.73  $K^2$ . For  $T_g$ , the  $R^2$  is 0.89, with a  $MAE$  of 22.72 K and a  $MSE$  of 3786.02  $K^2$ . As for  $T_x$ , the  $R^2$  is 0.98, with a  $MAE$  of 12.19 K and a  $MSE$  of 413.13  $K^2$ . Table 3 presents the calculation formulas and corresponding prediction accuracies of the previous studies related to the GFA parameters. Two categories of GFA parameters were identified: the direct use of  $D_{max}$  and mixed mathematical expressions involving  $T_l$ ,  $T_g$ , and  $T_x$ . However, both approaches yield relatively low prediction accuracies. In contrast, this study achieved  $R^2$  values greater than 0.91 for  $T_l$ ,  $T_g$ , and  $T_x$  predictions. Therefore, utilizing this study's method to predict  $T_l$ ,  $T_g$ , and  $T_x$  followed by corresponding parameter calculations can effectively enhance the prediction accuracy.

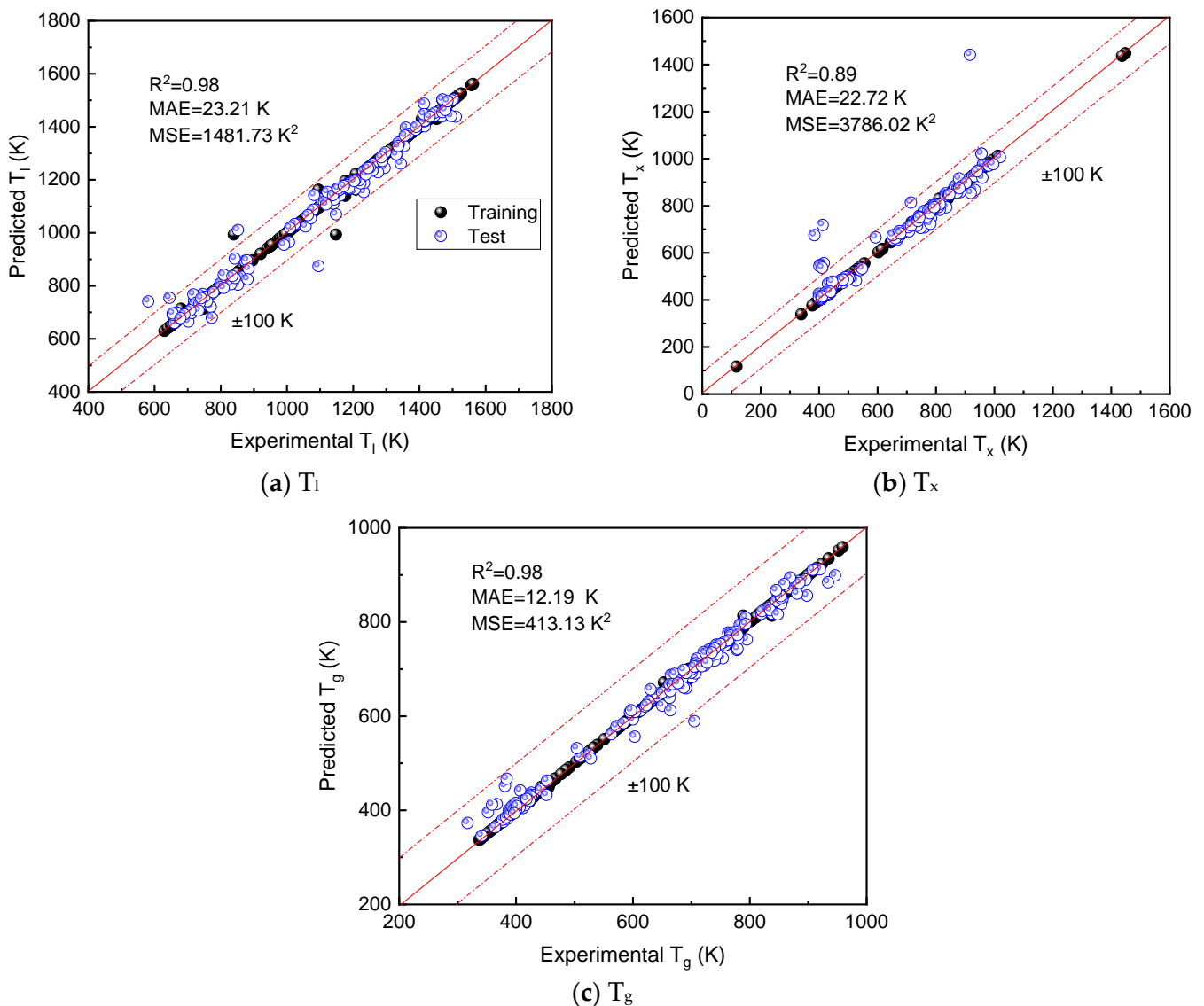
**Table 3.** GFA parameters and corresponding prediction accuracy.

No	Criteria	Formula	R <sup>2</sup>	Ref.
1	T <sub>rg</sub>	$T_{rg} = T_g/T_1$	0.24168	[80]
2	δ	$\delta = T_x/(T_1 - T_g)$	0.34638	[81]
3	β <sub>r</sub>	$\beta_r = (T_x T_g)/(T_1 - T_x)^2$	0.45692	[82]
4	w	$w = T_1(T_1 + T_x)/[T_x(T_1 - T_x)]$	0.48693	[83]
5	ΔT <sub>x</sub>	$\Delta T_x = T_x - T_g$	0.44805	[16]
6	γ	$\gamma = T_x/(T_g + T_1)$	0.50375	[17]
7	β <sub>l</sub>	$\beta_l = T_x/T_g + T_g/T_1$	0.51967	[84]
8	γ <sub>m</sub>	$\gamma_m = (2T_x - T_g)/T_1$	0.52861	[85]
9	v	$v = T_x T_g (T_x - T_g)/(T_1 - T_x)^3$	0.59435	[86]
10	w <sub>B</sub>	$w_B = (2T_x - T_g)/(T_1 + T_x)$	0.53842	[87]
11	γ <sub>c</sub>	$\gamma_c = (3T_x - 2T_g)/T_1$	0.55126	[88]
12	γ <sub>n</sub>	$(5T_x - 3T_g)/T_1$	0.26394	[89]
13	w <sub>l</sub>	$T_g/T_x - 2T_g/(T_g + T_1)$	0.24006	[24]
14	χ	$\chi = [(T_x - T_g)/(T_1 - T_x)][T_x/(T_1 - T_x)]^{1.47}$	0.60217	[90]
15	G <sub>p</sub>	$G_p = T_g(T_x - T_g)/(T_1 - T_x)^2$	0.5999	[6]
16	D <sub>max</sub>	-	0.70566	[9]
17	D <sub>max</sub>	-	0.795	[91]



**Figure 4.** The performance of T<sub>1</sub>, T<sub>x</sub>, and T<sub>g</sub> on different regression models of machine learning.





**Figure 5.** The comparison of predicted values and experimental values of  $T_l$ ,  $T_x$ , and  $T_g$  by ETR model.

In order to simplify the model and identify the key features that affect the GFA, it is necessary to perform feature selection on the input features of the model. Figure 6 illustrates the process of feature selection for three representative parameters:  $T_l$ ,  $T_x$ , and  $T_g$ . Each parameter's input features are subjected to variance filtering, correlation filtering, embedded method filtering, recursive filtering, and exhaustive filtering. This gradual process of dimensionality reduction ensures that the model's accuracy is not compromised. The changes in feature dimensionality throughout this process are depicted in Figure 7. After applying these dimensionality reduction methods, the input feature dimensions for all three parameters are reduced to 4, greatly simplifying the model. Additionally, the key features that significantly influence the GFA are identified and presented in Table 4. These features can be broadly categorized into two classes: basic properties, such as melting point, volume, column, and Mendeleev Number, as well as electronic characteristics, such as electronegativity, unfilled electron number, and magnetic moment.

Analyzing the selected features can provide insight into their impact on the target variable. After several decades of research on MGs, researchers have identified several principles that contribute to improving the GFA, including the Inoue empirical rules [16], the similar atom substitution principle [92], and the electron concentration principle [93,94]. According to the Inoue empirical rules, the formation of MGs with high GFA should

follow three principles: (1) the alloy system should consist of three or more elements; (2) the atomic size difference between the primary alloy components should exceed 14%; (3) and the mixing enthalpy between the primary composition elements in the alloy system should be negative. These rules indicate that the GFA is linked to the composition of elements, atomic size, and mixing enthalpy. On the other hand, the Miedema model [95,96] establishes relationships between the basic properties of elements (e.g., melting point, volume, column) and these parameters, providing partial insights into the GFA. The electron concentration principle proposes criteria for determining equivalent electron concentration, while electronic characteristics, such as electronegativity, unfilled electron number, and magnetic moment, can effectively explain variations in electron concentration, thereby influencing the GFA.

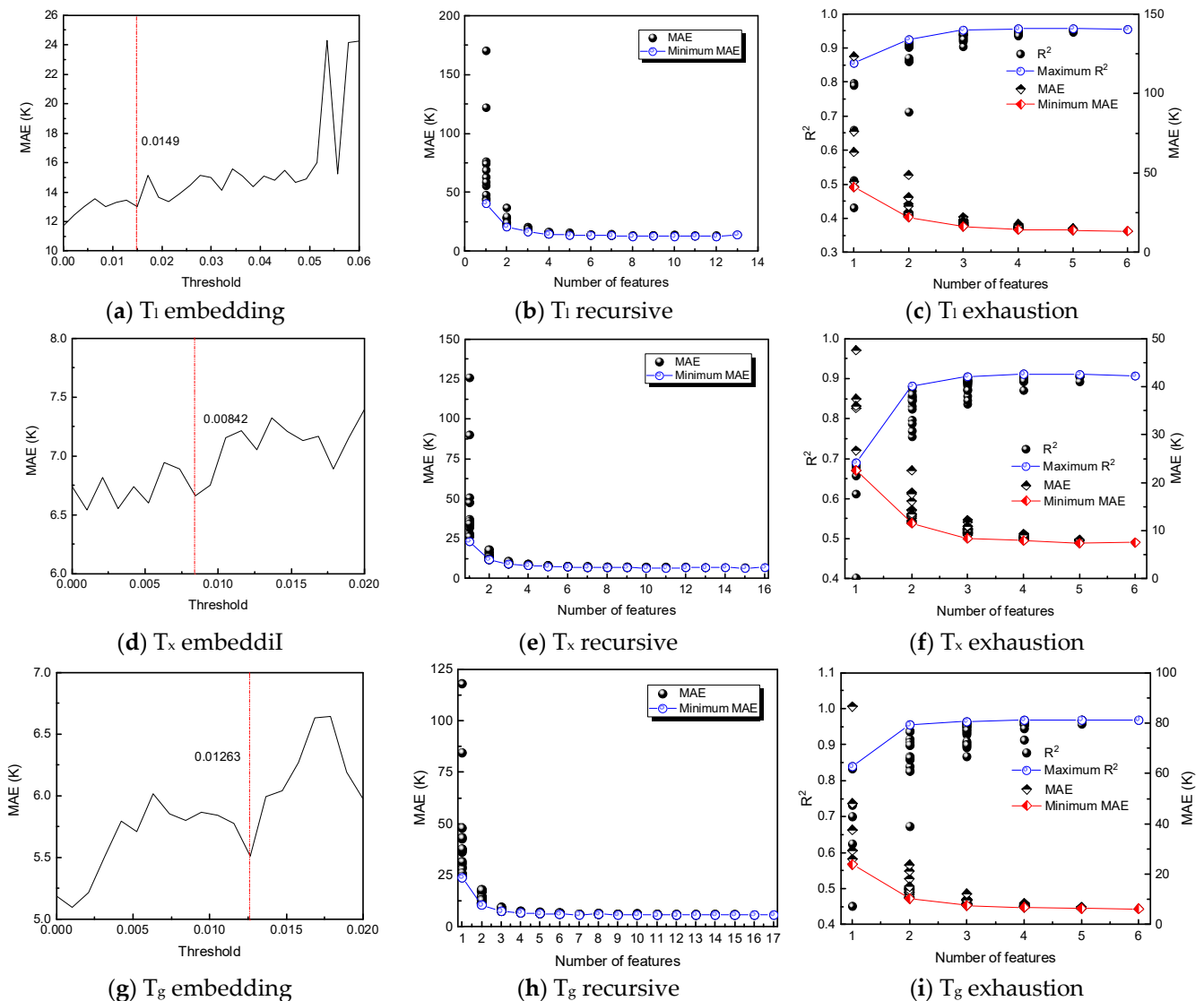
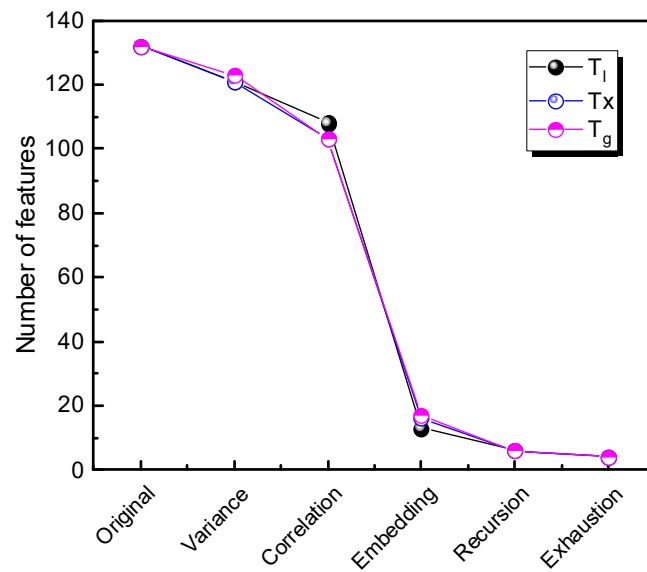


Figure 6. The feature screening process of  $T_1$ ,  $T_x$ , and  $T_g$ .



**Figure 7.** The change of feature number in the process of feature screening.

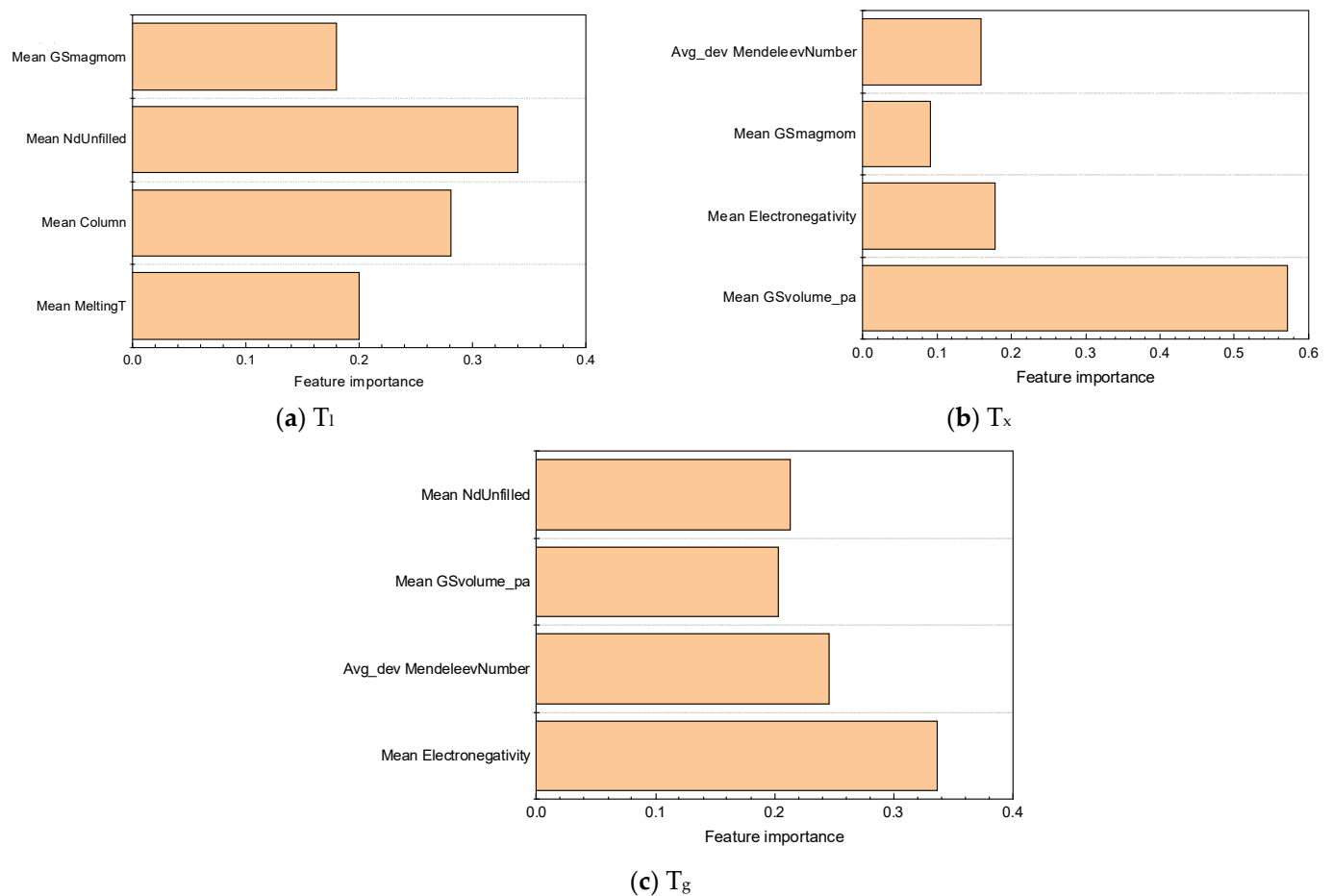
**Table 4.** The key features after screening.

	Key Feature	Description
T <sub>l</sub>	Mean GSvolume_pa	Mean atomic volume
	Mean Electronegativity	Average electronegativity of elements
	Mean Gsmagmom	Average magnetic moment of elements
	Avg_dev MendeleevNumber	Average deviation of Mendeleev number
T <sub>x</sub>	Mean Electronegativity	Average electronegativity of elements
	Avg_dev MendeleevNumber	Average deviation of Mendeleev number
	Mean GSvolume_pa	Mean atomic volume
	Mean NdUnfilled	Average orbital number of electrons in d layer that are not filled
T <sub>g</sub>	Mean MeltingT	Average melting point of elements
	Mean Column	Mean column of elements
	Mean Gsmagmom	Average magnetic moment of elements
	Mean NdUnfilled	Average orbital number of electrons in d layer that are not filled

### 3.3. Analysis of Key Features

#### (1) Feature importance analysis

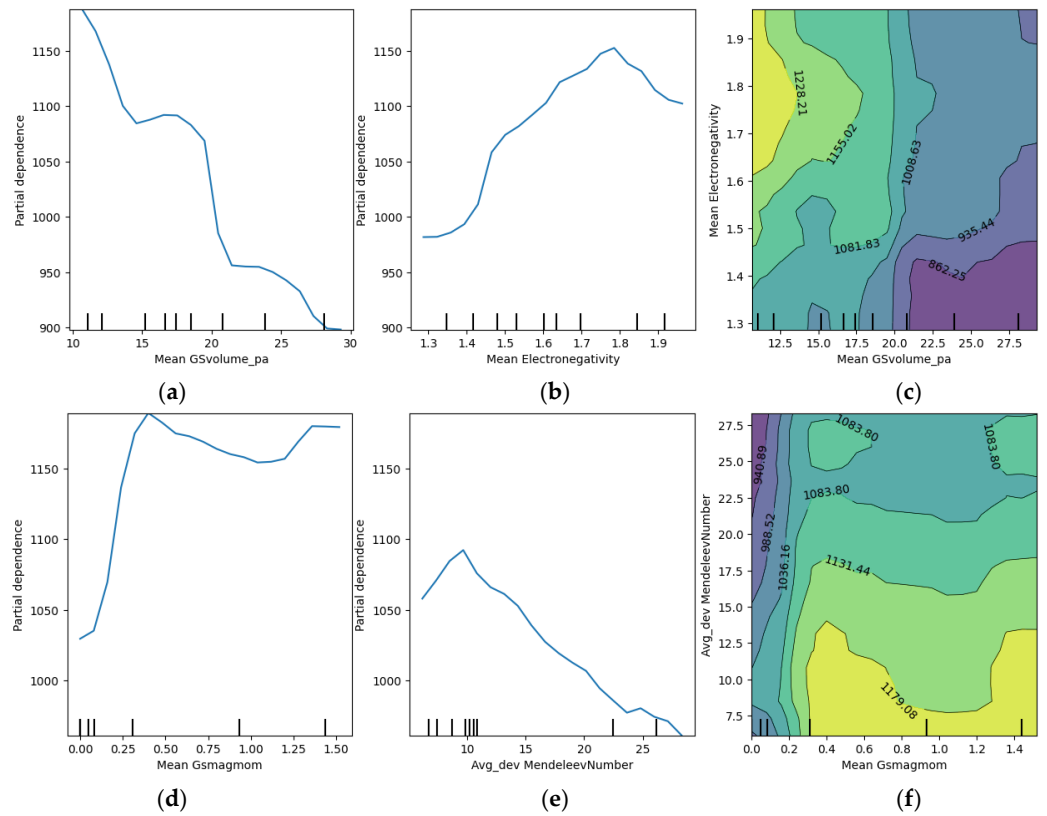
In order to comprehend the impact of features on the ultimate target value, an analysis was performed to determine the importance of each feature in the established prediction model, as illustrated in Figure 8. The results indicated that the average orbital number of electrons in d layer that are not filled, the average atomic volume, and the average electronegativity emerged as the most significant features for T<sub>g</sub>, T<sub>x</sub>, and T<sub>l</sub>, respectively. Consequently, when devising the composition of MGs, these features should be given particular consideration.



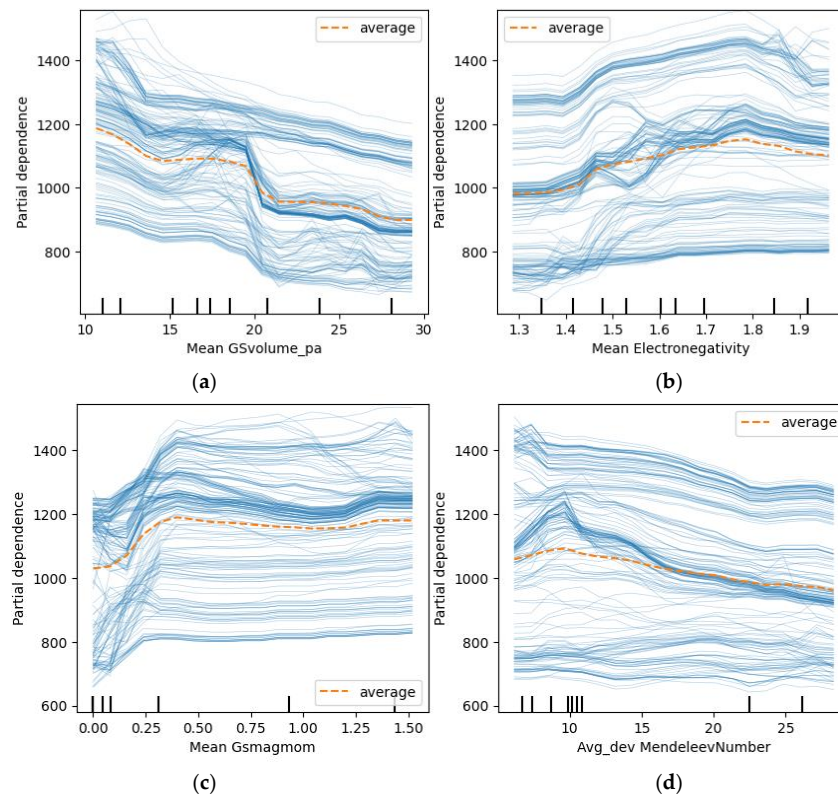
**Figure 8.** Rank of feature importance corresponding to  $T_g$ ,  $T_x$ , and  $T_1$ .

## (2) Analysis of PDP and ICE plots

In order to provide a more visually informative illustration of how these features impact the target value, we utilized PDP and ICE plots for feature analysis. The PDP plot showcases the influence of a single variable on the average target value. Figure 9 shows the PDP plot analysis of key features versus  $T_1$ . As Mean GSvolumepa increases,  $T_1$  consistently decreases, with a brief spike when Mean GSvolumepa reaches 20, followed by a continued decline. Conversely,  $T_1$  increases steadily as Mean electronegativity rises until it reaches 1.8, after which,  $T_1$  begins to decline. Figure 9c shows the combined effect of Mean GSvolumepa and Mean electronegativity on  $T_1$ . Specifically,  $T_1$  tends to be smaller when Mean GSvolumepa ranges from 25 to 28 and Mean electronegativity ranges from 1.3 to 1.4. Conversely,  $T_1$  tends to be larger when Mean GSvolumepa ranges from 12 to 13 and Mean electronegativity ranges from 1.6 to 1.9. Likewise, as Mean Gsmagmom increases,  $T_1$  gradually increases as well, with a temporary dip when Mean Gsmagmom falls between 0.25 and 1, followed by a continuous rise. On the other hand,  $T_1$  initially increases with the growth of Avg\_dev MendeleevNumber and then steadily decreases. Figure 9f shows the combined influence of Mean Gsmagmom and Avg\_dev MendeleevNumber on  $T_1$ . It reveals that  $T_1$  tends to be relatively small when Mean Gsmagmom ranges from 0 to 0.2 and Avg\_dev MendeleevNumber ranges from 22.5 to 27.5. Conversely,  $T_1$  tends to be larger when Minimum MeltingT ranges from 0.4 to 1.4 and Avg\_dev MendeleevNumber ranges from 7.5 to 12.5. Since the PDP plot solely showcases the impact of variables on the average target value and does not capture the variations in each sample's target value with respect to the features, ICE plots were additionally employed. Figure 10 shows ICE plot analysis of key features on  $T_1$ . While the target values differ across samples, they generally exhibit the same overall trend as the average target value.



**Figure 9.** PDP plot analysis of key features on  $T_1$ . (a) Mean GSvolum\_e\_pa. (b) Mean Electronegativity. (c) Mean GSvolum\_e\_pa and Mean Electronegativity. (d) Mean Gsmagmom. (e) Avg\_dev MendeleevNumb. (f) Mean Gsmagmom and Avg\_dev MendeleevNumber.

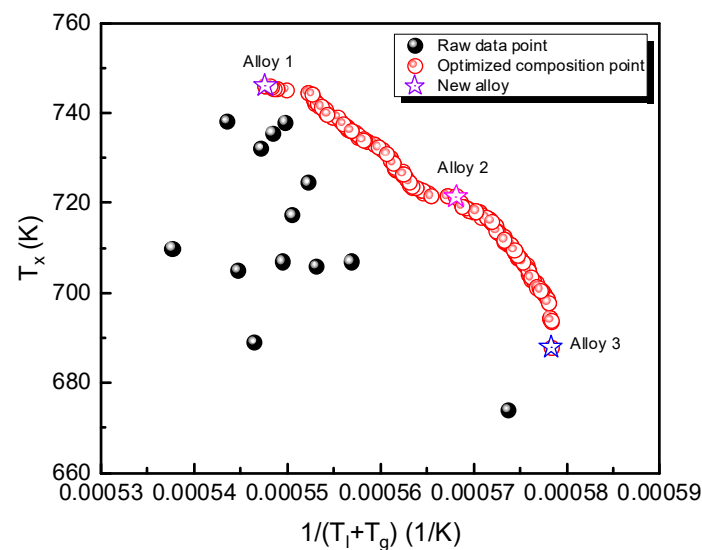


**Figure 10.** ICE plot analysis of key features on  $T_1$ . (a) Mean GSvolum\_e\_pa (b) Mean Electronegativity (c) Mean Gsmagmom (d) Avg\_dev MendeleevNumber.

### 3.4. Composition Optimization Design

Based on the depicted figure, it is evident that there are shared key features for  $T_l$ ,  $T_x$ , and  $T_g$ , and these features exert varying influences on the parameters. Taking the previously mentioned  $\gamma = T_x/(T_l + T_g)$  as an example, to optimize the GFA of MGs, it is essential to maximize  $\gamma$ . This, in turn, requires a compositional design approach that simultaneously maximizes  $T_x$  and  $1/(T_l + T_g)$ .

Taking the Zr-Cu-Ni-Al system as an example, the search range for each element is as follows: Cu: 5~20 wt.%, Ni: 5~20 wt.%, and Al: 5~10 wt.%. Each element is systematically varied within the compositional search space with a step size of 0.01 wt.%. The NSGA-II algorithm is utilized to simultaneously optimize  $T_x$  and  $1/(T_l + T_g)$  with the following specific parameter settings: the initial population size is 200, and the number of evolution generations is 300. It utilizes the polymutation mutation operator with a mutation rate of 0.02. The XOVR crossover operator is employed with a crossover rate of 0.9. The chromosomes are encoded using the RI encoding method. The primary objective is to identify a non-dominated set of compositions that maximize  $T_x$  and  $1/(T_l + T_g)$ , thereby facilitating the swift design of metallic glass compositions. Figure 11 visually presents the distribution of both the raw and optimized data for  $T_x$  and  $1/(T_l + T_g)$  within the Zr-Cu-Ni-Al system. The optimized data's Pareto front shifted towards the upper right corner compared to that of the raw data. This shift signifies that the application of a non-dominated genetic algorithm yielded superior composition designs. Three representative composition points are identified as Alloy 1: Zr61.46Cu8.96Ni19.98Al9.60, Alloy 2: Zr68.43Cu15.68Ni10.40Al5.49, and Alloy 3: Zr62.81Cu15.01Ni13.45Al8.73.



**Figure 11.** Raw data and optimized data distribution of Zr-Cu-Ni-Al metallic glasses.

The rapid design of alloys with high GFA is realized by machine learning, which effectively saves time and resources. However, the designed alloys need to be verified by experiments to prove the accuracy of the prediction model.

## 4. Conclusions

- (1) By developing a discriminative model based on alloy composition, the XGC model exhibits superior performance compared to the other three commonly employed classification models. With an impressive classification accuracy of 98%, it effectively determines GFA.
- (2) In the context of alloys capable of forming metallic glasses, the ETR algorithm is utilized to establish predictive models for  $T_l$ ,  $T_x$ , and  $T_g$ . The  $R^2$  values associated with these models all exceed 0.91, thereby demonstrating their exceptional predictive accuracy.

- (3) To enhance the simplicity of the models, various feature selection techniques, including variance, correlation, embedding, recursive, and exhaustive methods, are employed. These techniques enable the identification of crucial features for  $T_l$ ,  $T_x$ , and  $T_g$ . While ensuring the preservation of model accuracy, the dimensionality of the features is effectively reduced, resulting in the final selection of four key features for each property.
- (4) The interpretability analysis of the predictive models for  $T_l$ ,  $T_x$ , and  $T_g$  is performed by employing feature importance, PDP, and ICE. Through this comprehensive analysis, the influence patterns of each key feature on the target variables are uncovered, offering valuable insights for future alloy design. These findings serve as crucial reference directions for subsequent endeavors in alloy design.
- (5) In the case of the Zr-Cu-Al-Ni system, the GFA of MGs is evaluated through the parameter  $\gamma(T_x/(T_l + T_g))$ . The primary goal is to maximize  $\gamma$  by extensively exploring the compositional space of the Zr-Cu-Al-Ni system using a genetic algorithm. This innovative approach is aimed at enhancing the efficiency of MG development.

**Author Contributions:** Data curation, C.L. and X.W.; formal analysis, C.L. and W.C.; investigation, C.L., X.W., Y.H. and W.C.; conceptualization, H.S.; methodology, C.L., H.S. and Y.H.; software, C.L. and H.S.; validation, C.L. and Y.H.; funding acquisition, H.S. and Y.H.; supervision, H.S.; writing—original draft preparation, C.L.; writing—review and editing, H.S. All authors have read and agreed to the published version of the manuscript.

**Funding:** This research was funded by the National Key Research and Development Program of China (Grant No. 2021YFB3501502).

**Data Availability Statement:** The data presented in this study are available upon request from the corresponding author.

**Acknowledgments:** This work is supported by grants from the National Key Research and Development Program of China (Grant No. 2021YFB3501502). The authors are very grateful to the reviewers and editors for their valuable suggestions, which have helped improve the paper substantially.

**Conflicts of Interest:** The authors declare no conflict of interest.

## References

1. Egami, T.; Iwashita, T.; Dmowski, W. Mechanical Properties of Metallic Glasses. *Metals* **2013**, *3*, 77–113. [\[CrossRef\]](#)
2. Xu, D.; Lohwongwatana, B.; Duan, G.; Johnson, W.L.; Garland, C. Bulk metallic glass formation in binary Cu-rich alloy series— $\text{Cu}_{100-x}\text{Zr}_x$  ( $x = 34, 36, 38.2, 40$  at.%) and mechanical properties of bulk  $\text{Cu}_{64}\text{Zr}_{36}$  glass. *Acta Mater.* **2004**, *52*, 2621–2624. [\[CrossRef\]](#)
3. Wang, W.H. The elastic properties, elastic models and elastic perspectives of metallic glasses. *Prog. Mater. Sci.* **2012**, *57*, 487–656. [\[CrossRef\]](#)
4. Peter, W.H.; Buchanan, R.A.; Liu, C.T.; Liaw, P.K.; Morrison, M.L.; Horton, J.A.; Carmichael, C.A.; Wright, J.L. Localized corrosion behavior of a zirconium-based bulk metallic glass relative to its crystalline state. *Intermetallics* **2002**, *10*, 1157–1162. [\[CrossRef\]](#)
5. Jiao, Z.B.; Li, H.X.; Gao, J.E.; Wu, Y.; Lu, Z.P. Effects of alloying elements on glass formation, mechanical and soft-magnetic properties of Fe-based metallic glasses. *Intermetallics* **2011**, *19*, 1502–1508. [\[CrossRef\]](#)
6. Reddy, G.J.; Kandavalli, M.; Saboo, T.; Rao, A.K.P. Prediction of Glass Forming Ability of Bulk Metallic Glasses Using Machine Learning. *Integr. Mater. Manuf. Innov.* **2021**, *10*, 610–626. [\[CrossRef\]](#)
7. Xu, M.; Wang, J.; Sun, Y.; Zhu, S.; Zhang, T.; Guan, S. Prediction of glass-forming ability in ternary alloys based on machine learning method. *J. Non-Cryst. Solids* **2023**, *616*, 122476. [\[CrossRef\]](#)
8. Majid, A.; Ahsan, S.B.; Tariq, N.u.H. Modeling glass-forming ability of bulk metallic glasses using computational intelligent techniques. *Appl. Soft Comput.* **2015**, *28*, 569–578. [\[CrossRef\]](#)
9. Peng, L.; Long, Z.; Zhao, M. Determination of glass forming ability of bulk metallic glasses based on machine learning. *Comput. Mater. Sci.* **2021**, *195*, 110480. [\[CrossRef\]](#)
10. Xiong, J.; Shi, S.; Zhang, T. A machine-learning approach to predicting and understanding the properties of amorphous metallic alloys. *Mater. Des.* **2020**, *187*, 108378. [\[CrossRef\]](#)
11. Long, Z.; Liu, W.; Zhong, M.; Zhang, Y.; Zhao, M.; Liao, G.; Chen, Z. A new correlation between the characteristic temperature and glass-forming ability for bulk metallic glasses. *J. Therm. Anal. Calorim.* **2018**, *132*, 1645–1660. [\[CrossRef\]](#)
12. Mastropietro, D.G.; Moya, J.A. Design of Fe-based bulk metallic glasses for maximum amorphous diameter ( $D_{\text{max}}$ ) using machine learning models. *Comput. Mater. Sci.* **2021**, *188*, 110230. [\[CrossRef\]](#)

13. Lu, F.; Liang, Y.; Wang, X.; Gao, T.; Chen, Q.; Liu, Y.; Zhou, Y.; Yuan, Y.; Liu, Y. Prediction of amorphous forming ability based on artificial neural network and convolutional neural network. *Comput. Mater. Sci.* **2022**, *210*, 111464. [[CrossRef](#)]
14. Turnbull, D. Under what conditions can a glass be formed? *Contemp. Phys.* **1969**, *10*, 473–488. [[CrossRef](#)]
15. Chen, H.S. Thermodynamic considerations on the formation and stability of metallic glasses. *Scr. Metall.* **1974**, *22*, 1505–1511.
16. Inoue, A. Stabilization of metallic supercooled liquid and bulk amorphous alloys. *Acta Mater.* **2000**, *48*, 279–306. [[CrossRef](#)]
17. Lu, Z.P.; Liu, C.T. A new glass-forming ability criterion for bulk metallic glasses. *Acta Mater.* **2002**, *50*, 3501–3512. [[CrossRef](#)]
18. Ward, L.; O’Keeffe, S.C.; Stevick, J.; Jelbert, G.R.; Aykol, M.; Wolverton, C. A machine learning approach for engineering bulk metallic glass alloys. *Acta Mater.* **2018**, *159*, 102–111. [[CrossRef](#)]
19. Yu, J.Z. *Nonequilibrium Phase Diagrams of Ternary Amorphous Alloys*; Springer: Berlin/Heidelberg, Germany, 1997.
20. Huang, X.; Chang, C.; Chang, Z.; Wang, X.; Cao, Q.; Shen, B.; Inoue, A.; Jiang, J. Formation of bulk metallic glasses in the Fe–M–Y–B (M = transition metal) system. *J. Alloys Compd.* **2008**, *460*, 708–713. [[CrossRef](#)]
21. Kato, H.; Chen, H.S.; Inoue, A. Relationship between thermal expansion coefficient and glass transition temperature in metallic glasses. *Scr. Mater.* **2008**, *58*, 1106–1109. [[CrossRef](#)]
22. An, W.; Cai, A.; Li, J.; Luo, Y.; Li, T.; Xiong, X.; Liu, Y.; Pan, Y. Glass formation and non-isothermal crystallization of Zr<sub>62.5</sub>Al<sub>12.1</sub>Cu<sub>7.95</sub>Ni<sub>17.45</sub> bulk metallic glass. *J. Non-Cryst. Solids* **2009**, *355*, 1703–1706. [[CrossRef](#)]
23. González, S.; Figueroa, I.; Todd, I. Influence of minor alloying additions on the glass-forming ability of Mg–Ni–La bulk metallic glasses. *J. Alloys Compd.* **2009**, *484*, 612–618. [[CrossRef](#)]
24. Long, Z.; Wei, H.; Ding, Y.; Zhang, P.; Xie, G.; Inoue, A. A new criterion for predicting the glass-forming ability of bulk metallic glasses. *J. Alloys Compd.* **2009**, *475*, 207–219. [[CrossRef](#)]
25. Huang, X.; Chang, C.; Chang, Z.; Inoue, A.; Jiang, J. Glass forming ability, mechanical and magnetic properties in Fe–W–Y–B alloys. *Mater. Sci. Eng. A* **2010**, *527*, 1952–1956. [[CrossRef](#)]
26. Guo, S.-f.; Ye, S. Design of high strength Fe–(P, C)-based bulk metallic glasses with Nb addition. *Trans. Nonferrous Met. Soc. China* **2011**, *21*, 2433–2437. [[CrossRef](#)]
27. Hua, N.; Li, R.; Wang, H.; Wang, J.; Li, Y.; Zhang, T. Formation and mechanical properties of Ni-free Zr-based bulk metallic glasses. *J. Alloys Compd.* **2011**, *509*, S175–S178. [[CrossRef](#)]
28. Kucuk, I.; Aykol, M.; Uzun, O.; Yildirim, M.; Kabaer, M.; Duman, N.; Yilmaz, F.; Erturk, K.; Akdeniz, M.V.; Mekhrabov, A.O. Effect of (Mo, W) substitution for Nb on glass forming ability and magnetic properties of Fe–Co-based bulk amorphous alloys fabricated by centrifugal casting. *J. Alloys Compd.* **2011**, *509*, 2334–2337. [[CrossRef](#)]
29. Sheng, G.; Liu, C.T. Phase stability in high entropy alloys: Formation of solid-solution phase or amorphous phase. *Prog. Nat. Sci. Mater. Int.* **2011**, *21*, 433–446.
30. Samavatian, M.; Gholamipour, R.; Samavatian, V. Discovery of novel quaternary bulk metallic glasses using a developed correlation-based neural network approach. *Comput. Mater. Sci.* **2021**, *186*, 110025. [[CrossRef](#)]
31. Deshmukh, A.A.; Khond, A.A.; Bhatt, J.G.; Palikundwar, U.A. Understanding the role of Er on glass forming ability parameters and critical cooling rate in Fe–based multicomponent bulk metallic glasses. *J. Alloys Compd.* **2020**, *819*, 152938. [[CrossRef](#)]
32. Li, B.; Sun, W.C.; Qi, H.N.; Lv, J.W.; Wang, F.L.; Ma, M.Z.; Zhang, X.Y. Effects of Ag substitution for Fe on glass-forming ability, crystallization kinetics, and mechanical properties of Ni-free Zr–Cu–Al–Fe bulk metallic glasses. *J. Alloys Compd.* **2020**, *827*, 154385. [[CrossRef](#)]
33. Jia, H.; Xie, X.; Zhao, L.; Wang, J.; Gao, Y.; Dahmen, K.A.; Li, W.; Liaw, P.K.; Ma, C. Effects of similar-element-substitution on the glass-forming ability and mechanical behaviors of Ti–Cu–Zr–Pd bulk metallic glasses. *J. Mater. Res. Technol.* **2018**, *7*, 261–269. [[CrossRef](#)]
34. Hu, F.; Yuan, C.; Luo, Q.; Yang, W.; Shen, B. Effects of heavy rare-earth addition on glass-forming ability, thermal, magnetic, and mechanical properties of Fe–RE–B–Nb (RE = Dy, Ho, Er or Tm) bulk metallic glass. *J. Non-Cryst. Solids* **2019**, *525*, 119681. [[CrossRef](#)]
35. Hu, F.; Luo, Q.; Shen, B. Thermal, magnetic and magnetocaloric properties of FeErNbB metallic glasses with high glass-forming ability. *J. Non-Cryst. Solids* **2019**, *512*, 184–188. [[CrossRef](#)]
36. Hasani, S.; Rezaei-Shahreza, P.; Seifoddini, A.; Hakimi, M. Enhanced glass forming ability, mechanical, and magnetic properties of Fe<sub>41</sub>Co<sub>7</sub>Cr<sub>15</sub>Mo<sub>14</sub>Y<sub>2</sub>C<sub>15</sub>B<sub>6</sub> bulk metallic glass with minor addition of Cu. *J. Non-Cryst. Solids* **2018**, *497*, 40–44. [[CrossRef](#)]
37. Gu, J.-L.; Shao, Y.; Yao, K.-F. The novel Ti-based metallic glass with excellent glass forming ability and an elastic constant dependent glass forming criterion. *Materialia* **2019**, *8*, 100433. [[CrossRef](#)]
38. Ge, J.; He, H.; Zhou, J.; Lu, C.; Dong, W.; Liu, S.; Lan, S.; Wu, Z.; Wang, A.; Wang, L.; et al. In-situ scattering study of a liquid-liquid phase transition in Fe–B–Nb–Y supercooled liquids and its correlation with glass-forming ability. *J. Alloys Compd.* **2019**, *787*, 831–839. [[CrossRef](#)]
39. Zhu, J.; Wang, C.; Han, J.; Yang, S.; Xie, G.; Jiang, H.; Chen, Y.; Liu, X. Formation of Zr-based bulk metallic glass with large amount of yttrium addition. *Intermetallics* **2018**, *92*, 55–61. [[CrossRef](#)]
40. Yang, G.; Lian, J.; Wang, R.; Wu, N. Similar atom substitution effect on the glass forming ability in (La Ce) Al–(Ni Co) bulk metallic glasses using electron structure guiding. *J. Alloys Compd.* **2019**, *786*, 250–256. [[CrossRef](#)]
41. Yang, Y.J.; Cheng, B.Y.; Lv, J.W.; Li, B.; Ma, M.Z.; Zhang, X.Y.; Li, G.; Liu, R.P. Effect of Ag substitution for Ti on glass-forming ability, thermal stability and mechanical properties of Zr-based bulk metallic glasses. *Mater. Sci. Eng. A* **2019**, *746*, 229–238. [[CrossRef](#)]



42. Wada, T.; Jiang, J.; Yubuta, K.; Kato, H.; Takeuchi, A. Septenary Zr–Hf–Ti–Al–Co–Ni–Cu high-entropy bulk metallic glasses with centimeter-scale glass-forming ability. *Materialia* **2019**, *7*, 100372. [[CrossRef](#)]
43. Dong, Q.; Pan, Y.J.; Tan, J.; Qin, X.M.; Li, C.J.; Gao, P.; Feng, Z.X.; Calin, M.; Eckert, J. A comparative study of glass-forming ability, crystallization kinetics and mechanical properties of  $Zr_{55}Co_{25}Al_{20}$  and  $Zr_{52}Co_{25}Al_{23}$  bulk metallic glasses. *J. Alloys Compd.* **2019**, *785*, 422–428. [[CrossRef](#)]
44. Xue, L.; Li, J.; Yang, W.; Yuan, C.; Shen, B. Effect of Fe substitution on magnetocaloric effects and glass-forming ability in Gd-based metallic glasses. *Intermetallics* **2018**, *93*, 67–71. [[CrossRef](#)]
45. Cao, D.; Wu, Y.; Liu, X.J.; Wang, H.; Wang, X.Z.; Lu, Z.P. Enhancement of glass-forming ability and plasticity via alloying the elements having positive heat of mixing with Cu in  $Cu_{48}Zr_{48}Al_4$  bulk metallic glass. *J. Alloys Compd.* **2019**, *777*, 382–391. [[CrossRef](#)]
46. Malekan, M.; Rashidi, R.; Shabestari, S.G. Mechanical properties and crystallization kinetics of Er-containing Cu–Zr–Al bulk metallic glasses with excellent glass forming ability. *Vacuum* **2020**, *174*, 109223. [[CrossRef](#)]
47. Saini, S.; Srivastava, A.P.; Neogy, S. The effect of Ag addition on the crystallization kinetics and glass forming ability of Zr–(CuAg)–Al bulk metallic glass. *J. Alloys Compd.* **2019**, *772*, 961–967. [[CrossRef](#)]
48. Liang, D.-d.; Wei, X.; Chang, C.; Li, J.; Wang, X.; Shen, J. Effect of W addition on the glass forming ability and mechanical properties of Fe-based metallic glass. *J. Alloys Compd.* **2018**, *731*, 1146–1150. [[CrossRef](#)]
49. Rahvard, M.M.; Tamizifar, M.; Boutorabi, S.M.A. Zr–Co(Cu)–Al bulk metallic glasses with optimal glass-forming ability and their compressive properties. *Trans. Nonferrous Met. Soc. China* **2018**, *28*, 1543–1552. [[CrossRef](#)]
50. Zhang, T.; Long, Z.; Peng, L.; Li, Z. Prediction of glass forming ability of bulk metallic glasses based on convolutional neural network. *J. Non-Cryst. Solids* **2022**, *595*, 121846. [[CrossRef](#)]
51. Ma, S.; Ran, Y.; Liang, X.; Jiang, L.; Li, Y.; Wang, X.; Yao, M.; Zhang, W. Unveiling the role of Y content in glass-forming ability and soft magnetic properties of Co–Y–B metallic glasses by experiment and ab initio molecular dynamics simulations. *J. Alloys Compd.* **2022**, *902*, 163637. [[CrossRef](#)]
52. Malekan, M.; Rashidi, R.; Shabestari, S.G.; Eckert, J. Thermodynamic and kinetic interpretation of the glass-forming ability of Y-containing Cu–Zr–Al bulk metallic glasses. *J. Non-Cryst. Solids* **2022**, *576*, 121266. [[CrossRef](#)]
53. Wen, S.; Dai, C.; Mao, W.; Zhao, Y.; Han, G.; Wang, X. Effects of Ag and Co microalloying on glass-forming abilities and plasticity of Cu–Zr–Al based bulk metallic glasses. *Mater. Des.* **2022**, *220*, 110896. [[CrossRef](#)]
54. Ren, J.; Li, Y.; Liang, X.; Kato, H.; Zhang, W. Role of Fe substitution for Co on thermal stability and glass-forming ability of soft magnetic Co-based Co–Fe–B–P–C metallic glasses. *Intermetallics* **2022**, *147*, 107598. [[CrossRef](#)]
55. Ma, S.; Zhang, J.; Wang, X.; Umetsu, R.Y.; Jiang, L.; Zhang, W.; Yao, M. Structural Origins for Enhanced Thermal Stability and Glass-Forming Ability of Co–B Metallic Glasses with Y and Nb Addition. *Acta Metall. Sin. (Engl. Lett.)* **2023**, *36*, 962–972. [[CrossRef](#)]
56. Zhu, J.; Gao, W.; Cheng, S.; Liu, X.; Yang, X.; Tian, J.; Ma, J.; Shen, J. Improving the glass forming ability and plasticity of ZrCuNiAlTi metallic glass by substituting Zr with Sc. *J. Alloys Compd.* **2022**, *909*, 164679. [[CrossRef](#)]
57. Ohashi, Y.; Wada, T.; Kato, H. High-entropy design and its influence on glass-forming ability in Zr–Cu-based metallic glass. *J. Alloys Compd.* **2022**, *915*, 165366. [[CrossRef](#)]
58. Liu, X.W.; Long, Z.L.; Zhang, W.; Yang, L.M. Key feature space for predicting the glass-forming ability of amorphous alloys revealed by gradient boosted decision trees model. *J. Alloys Compd.* **2022**, *901*, 163606. [[CrossRef](#)]
59. Wang, Y.; Wang, A.; Li, H.; Zhang, H.; Zhu, Z. The effect of minor alloying on the glass forming ability and crystallization reaction of  $Ti_{32.8}Zr_{30.2}Cu_9M_{5.3}Be_{22.7}$  (M = Fe, Co, and Ni) bulk metallic glass. *J. Mater. Res. Technol.* **2022**, *18*, 3035–3043. [[CrossRef](#)]
60. Zhang, S.; Wei, C.; Shi, Z.; Zhang, H.; Ma, M. Effect of Fe addition on the glass-forming ability, stability, and mechanical properties of  $Zr_{50}Cu_{34}$ –Fe  $Al_8Ag_8$  metallic glasses. *J. Alloys Compd.* **2022**, *929*, 167334. [[CrossRef](#)]
61. Zhou, Y.; Zhao, L.; Qu, Y.; Hu, L.; Qi, L.; Qu, F.; He, S.; Liu, X. Effect of Yttrium Doping on Glass-Forming Ability, Thermal Stability, and Corrosion Resistance of  $Zr_{50.7}Cu_{28}Ni_9Al_{12.3}$  Bulk Metallic Glass. *Metals* **2023**, *13*, 521. [[CrossRef](#)]
62. Ma, X.; Li, Q.; Xie, L.; Chang, C.; Li, H. Effect of Ni addition on the properties of CoMoPB bulk metallic glasses. *J. Non-Cryst. Solids* **2022**, *587*, 121573. [[CrossRef](#)]
63. Lu, S.; Li, X.; Liang, X.; He, J.; Shao, W.; Li, K.; Chen, J. Effect of Ho Addition on the Glass-Forming Ability and Crystallization Behaviors of  $Zr_{54}Cu_{29}Al_{10}Ni_7$  Bulk Metallic Glass. *Metals* **2022**, *15*, 2516. [[CrossRef](#)] [[PubMed](#)]
64. Huang, Z.; Tao, P.; Long, Z.; Xiong, Z.; Zhu, X.; Xu, X.; Huang, Z.; Deng, H.; Lin, H.; Li, W.; et al. Effect of Ti addition on mechanical properties of Zr-based bulk metallic glasses. *J. Non-Cryst. Solids* **2023**, *601*, 122075. [[CrossRef](#)]
65. Peng, J.; Tang, B.; Wang, Q.; Bai, C.; Wu, Y.; Chen, Q.; Li, D.; Ding, D.; Xia, L.; Guo, X.; et al. Effect of heavy rare-earth (Dy, Tb, Gd) addition on the glass-forming ability and magneto-caloric properties of  $Fe_{89}Zr_7B_4$  amorphous alloy. *J. Alloys Compd.* **2022**, *925*, 166707. [[CrossRef](#)]
66. Diao, Y.; Yan, L.; Gao, K. Improvement of the machine learning-based corrosion rate prediction model through the optimization of input features. *Mater. Des.* **2021**, *198*, 109326. [[CrossRef](#)]
67. Lix, L.M.; Keselman, J.C.; Keselman, H.J. Consequences of Assumption Violations Revisited: A Quantitative Review of Alternatives to the One-Way Analysis of Variance F Test. *Rev. Educ. Res.* **1996**, *66*, 579–619.
68. Sekeh, S.Y.; Hero, A.O. Feature Selection For Mutliti-Labeled Variables Via Dependency Maximization. In Proceedings of the IEEE International Conference on Acoustics, Speech, Signal Processing, Brighton, UK, 12–17 May 2019; pp. 3127–3131.

69. Louw, N.; Steel, S.J. Variable selection in kernel Fisher discriminant analysis by means of recursive feature elimination. *Comput. Stat. Data Anal.* **2005**, *51*, 2043–2055. [[CrossRef](#)]
70. Wang, L.; Wang, Y.; Chang, Q. Feature Selection Methods for Big Data Bioinformatics: A Survey from the Search Perspective. *Methods* **2016**, *111*, 21–31. [[CrossRef](#)]
71. Islam, N.; Huang, W.; Zhuang, H.L. Machine learning for phase selection in multi-principal element alloys. *Comput. Mater. Sci.* **2018**, *150*, 230–235. [[CrossRef](#)]
72. Liu, Y.; Hou, T.; Yan, Z.; Yu, T.; Duan, J.; Xiao, Y.; Wu, K. The effect of element characteristics on bainite transformation start temperature using a machine learning approach. *J. Mater. Sci.* **2023**, *58*, 443–456. [[CrossRef](#)]
73. Meredig, B.; Agrawal, A.; Kirklin, S.; Saal, J.E.; Doak, J.W.; Thompson, A.; Zhang, K.; Choudhary, A.; Wolverton, C. Combinatorial screening for new materials in unconstrained composition space with machine learning. *Phys. Rev. B* **2014**, *89*, 094104. [[CrossRef](#)]
74. Friedman, J.H. Greedy Function Approximation: A Gradient Boosting Machine. *Ann. Stat.* **2001**, *29*, 1189–1232. [[CrossRef](#)]
75. Goldstein, A.; Kapelner, A.; Bleich, J.; Pitkin, E. Peeking Inside the Black Box: Visualizing Statistical Learning With Plots of Individual Conditional Expectation. *J. Comput. Graph. Stat.* **2015**, *24*, 44–65. [[CrossRef](#)]
76. Seibold, H.; Zeileis, A.; Hothorn, T. Model-Based Recursive Partitioning for Subgroup Analyses. *Int. Stat. Rev.* **2016**, *12*, 45–63. [[CrossRef](#)] [[PubMed](#)]
77. Yan, Z.; Li, L.; Cheng, L.; Chen, X.; Wu, K. New insight in predicting martensite start temperature in steels. *J. Mater. Sci.* **2022**, *57*, 11392–11410. [[CrossRef](#)]
78. Chand, S.; Wagner, M. Evolutionary many-objective optimization: A quick-start guide. *Surv. Oper. Res. Manag. Sci.* **2015**, *20*, 35–42. [[CrossRef](#)]
79. Liu, C.; Wang, X.; Cai, W.; Yang, J.; Su, H. Optimal Design of the Austenitic Stainless-Steel Composition Based on Machine Learning and Genetic Algorithm. *Materials* **2023**, *16*, 5633. [[CrossRef](#)]
80. Lu, Z.P.; Tan, H.; Ng, S.C.; Li, Y. The correlation between reduced glass transition temperature and glass forming ability of bulk metallic glasses. *Scr. Mater.* **2000**, *42*, 667–673. [[CrossRef](#)]
81. Chen, Q.; Shen, J.; Zhang, D.; Fan, H.; Sun, J.; McCartney, D.G. A new criterion for evaluating the glass-forming ability of bulk metallic glasses. *Mater. Sci. Eng. A* **2006**, *433*, 155–160. [[CrossRef](#)]
82. Yuan, Z.-Z.; Bao, S.-L.; Lu, Y.; Zhang, D.-P.; Yao, L. A new criterion for evaluating the glass-forming ability of bulk glass forming alloys. *J. Alloys Compd.* **2008**, *459*, 251–260. [[CrossRef](#)]
83. Ji, X.-L.; Pan, Y. A thermodynamic approach to assess glass-forming ability of bulk metallic glasses. *Trans. Nonferrous Met. Soc. China* **2009**, *19*, 1271–1279. [[CrossRef](#)]
84. Mondal, K.; Murty, B.S. On the parameters to assess the glass forming ability of liquids. *J. Non-Cryst. Solids* **2005**, *351*, 1366–1371. [[CrossRef](#)]
85. Du, X.H.; Huang, J.C.; Liu, C.T.; Lu, Z.P. New criterion of glass forming ability for bulk metallic glasses. *J. Appl. Phys.* **2007**, *101*, 086108. [[CrossRef](#)]
86. Błyskun, P.; Maj, P.; Kowalczyk, M.; Latuch, J.; Kulik, T. Relation of various GFA indicators to the critical diameter of Zr-based BMGs. *J. Alloys Compd.* **2015**, *625*, 13–17. [[CrossRef](#)]
87. Deng, R.; Long, Z.; Peng, L.; Kuang, D.; Ren, B. A new mathematical expression for the relation between characteristic temperature and glass-forming ability of metallic glasses. *J. Non-Cryst. Solids* **2020**, *533*, 119829. [[CrossRef](#)]
88. Guo, S.; Liu, C.T. New glass forming ability criterion derived from cooling consideration. *Intermetallics* **2010**, *18*, 2065–2068. [[CrossRef](#)]
89. Xiong, J.; Zhang, T.-Y.; Shi, S.-Q. Machine learning prediction of elastic properties and glass-forming ability of bulk metallic glasses. *MRS Commun.* **2019**, *9*, 576–585. [[CrossRef](#)]
90. Tripathi, M.K.; Ganguly, S.; Dey, P.; Chattopadhyay, P.P. Evolution of glass forming ability indicator by genetic programming. *Comput. Mater. Sci.* **2016**, *118*, 56–65. [[CrossRef](#)]
91. Long, T.; Long, Z.; Pang, B.; Li, Z.; Liu, X. Overcoming the challenge of the data imbalance for prediction of the glass forming ability in bulk metallic glasses. *Mater. Today Commun.* **2023**, *35*, 105610. [[CrossRef](#)]
92. Greer, A.L. Confusion by design. *Nature* **1993**, *366*, 303–304. [[CrossRef](#)]
93. Qiang, J.-B.; Wang, D.-H.; Bao, C.-M.; Wang, Y.-M.; Xu, W.-P.; Song, M.-L.; Dong, C. Formation rule for Al-based ternary quasi-crystals: Example of Al–Ni–Fe decagonal phase. *J. Mater. Res.* **2001**, *16*, 2653–2660. [[CrossRef](#)]
94. Wang, Y.M.; Qiang, J.B.; Wong, C.H.; Shek, C.H.; Dong, C. Composition rule of bulk metallic glasses and quasicrystals using electron concentration criterion. *J. Mater. Res.* **2003**, *18*, 642–648. [[CrossRef](#)]
95. Boer, F.R.d.; Mattens, W.C.M.; Boom, R.; Miedema, A.R.; Niessen, A.K. *Cohesion in Metals. Transition Metal Alloys*; North-Holland: Amsterdam, The Netherlands, 1988; Volume 1.
96. Miedema, A.R.; de Châtel, P.F.; de Boer, F.R. Cohesion in alloys—Fundamentals of a semi-empirical model. *Physica B+C* **1980**, *100*, 1–28. [[CrossRef](#)]

**Disclaimer/Publisher’s Note:** The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.